

## Lecture 24: 12.07.05 Flory-Huggins Theory

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**Reading:** Dill and Bromberg Ch. 31 'Polymer Solutions,' pp. 593-605.

### Supplementary Reading:

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### Announcements:

**Final quiz:** Tuesday, Dec. 20 1:30pm-4:30pm DuPont  
 Same format and length as first 2 quizzes; you will have 3 hrs but quiz should only take 2 hrs  
 Equation sheet is allowed, as in prior quizzes

**Review sessions (tentatively):** Thermo/Stat Mech review: Friday, Dec. 16, 5-7pm (normal recitation room)  
 Structure review: Sunday, Dec. 18, time TBD (normal recitation room)

**Thermo coverage on Quiz 3:** lectures 17-24

PROF. IRVING EXTRA OFFICE HRS: (8-425)

FRI, DEC. 16: 9-11am

MON, DEC. 19: 10am-NOON

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**Last time**


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**Lattice Models of Solutions**

- Lattice models are a form of **COARSE GRAINING**, where only the most important molecular details of a system are retained in a statistical mechanical model. This is a powerful approach to create stat mech models where meaningful predictions can be made for complex materials. We began a derivation of the free energy of mixing for a polymer solution based on **Flory-Huggins theory**:

 $n_p$  = number polymer chains

 $N$  = number of segments per chain

 $n_s$  = number of solvent molecules

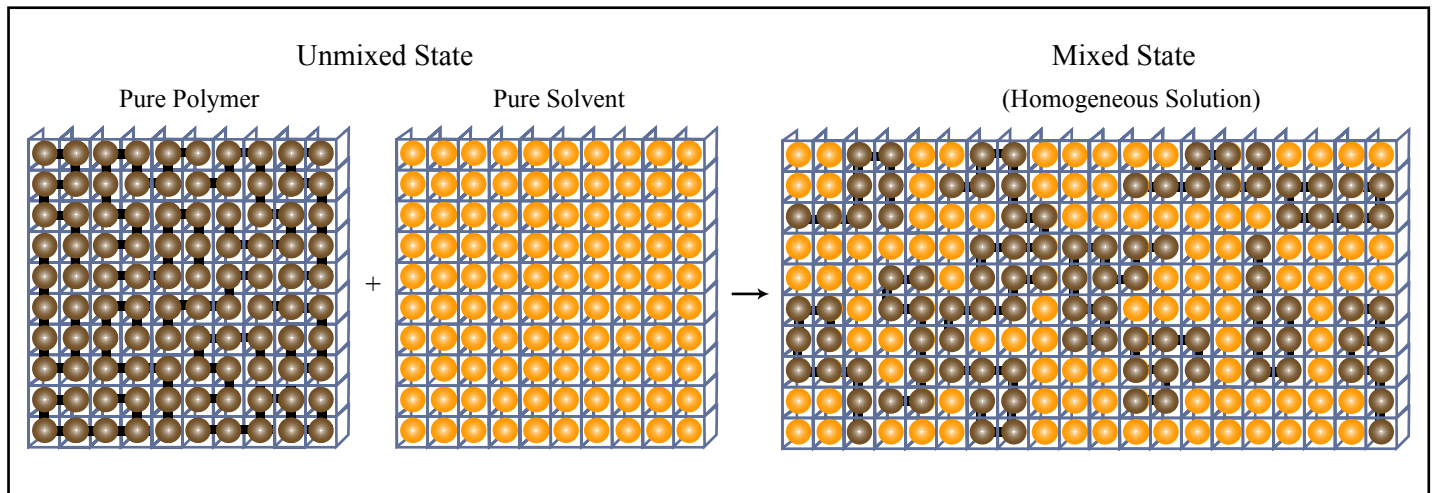
 $M$  = total lattice sites =  $n_s + Nn_p$ 


Figure by MIT OCW.

volume fraction of solvent:

$$\phi_p = \frac{V_p}{V_p + V_s} = \frac{Nn_p}{M}$$

volume fraction of solvent:

$$\phi_s = \frac{V_s}{V_p + V_s} = \frac{n_s}{M}$$

- Assumptions:**

- Volumes of one solvent molecule and one segment of the polymer chain are equal to one lattice site.
- Molecules mix randomly. ← *MEAN FIELD!*
- Only contribution to possible states are translational configurations ( $q = q_{\text{translation}}$ ).
- Molecules of a given type are indistinguishable.

- To predict the thermodynamic behavior of a polymer solution, we want to derive an expression for the free energy of mixing from the molecular states of the system:

- For a closed system at constant temperature and pressure:

$$\Delta G^{mix} = \Delta H^{mix} - T\Delta S^{mix} \quad \text{WHAT IS } W?$$

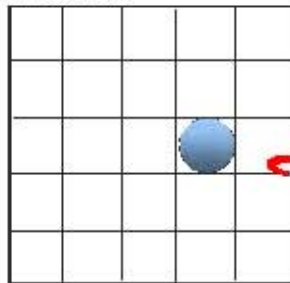
$$\Delta S = k_B \ln W$$

Entropy of mixing in the Flory-Huggins model

CONFIGURATIONS OF A SINGLE CHAIN

- How do we calculate  $W$ , the number of unique states in the polymer system? We start by looking at the number of ways to place a **single polymer chain** on the lattice,  $v_1$ :
  - What are the number of conformations for first bead?

M total sites



$v_1 = \text{\# CONFIGURATIONS FOR 1 POLYMER CHAIN} = (M) \dots$

M POSSIBLE SITES

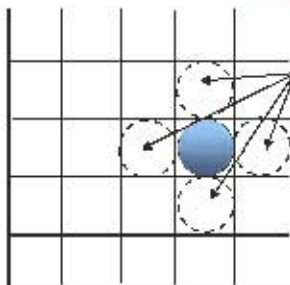
- With the first bead placed on the lattice, what is the number of possible locations for the second segment of the chain?

$v_1 = (M) (z) \dots$   
 1st 2nd

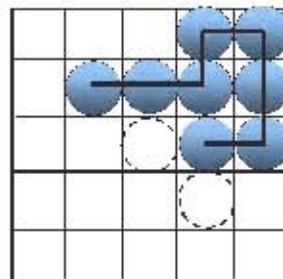
FRACTION OF AVAIL. SITES =  $\frac{M-1}{M}$

COORDINATION  $z$

$v_1 = M \left( z \left( \frac{M-1}{M} \right) \right) \dots$

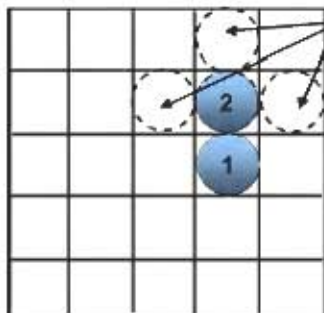


Z neighboring sites



Excluded volume: Number of available neighbor sites is reduced as chains fill up the lattice

- Moving on to placement of the third segment of the chain:



Z-1 available neighboring sites

$v_1 = (M) \left( z \left( \frac{M-1}{M} \right) \right) \left( (z-1) \left( \frac{M-2}{M} \right) \right) \dots$   
 1st 2nd 3rd

- We repeat this process to place all N segments of the chain on the lattice, and arrive at  $v_1$ , the **total number of configurations for a single chain**:

$$v_1 = (M) \left[ z \binom{M-1}{m} \right] \left[ (z-1) \binom{M-2}{m} \right] \left[ (z-1) \binom{M-3}{m} \right] \dots \left[ (z-1) \binom{M-(N-1)}{m} \right]$$

$$v_1 \approx \left( \frac{z-1}{m} \right)^{N-1} \frac{M!}{(M-N)!}$$

**COUNTING CONFIGURATIONS FOR A COLLECTION OF CHAINS**

- We can follow the same procedure used for a single chain to obtain the number of configurations possible for an entire set of  $n_p$  chains. We start by placing the **FIRST SEGMENT OF ALL  $n_p$  CHAINS**. The number of configurations for the first segment of all  $n_p$  chains is  $v_{first}$ :

$$v_{first} = M(M-1)(M-2)(M-3)\dots(M-(n_p-1)) = \frac{M!}{(M-n_p)!}$$

- The number of configurations for the (N - 1) remaining segments of all  $n_p$  chains is  $v_{subsequent}$ :

$$v_{subsequent} = \left[ (z-1) \binom{M-n_p}{M} \right] \left[ (z-1) \binom{M-n_p-1}{M} \right] \dots \left[ (z-1) \binom{M-2n_p}{M} \right] \dots \left[ (z-1) \binom{M-N(n_p-1)-(N-1)}{M} \right]$$

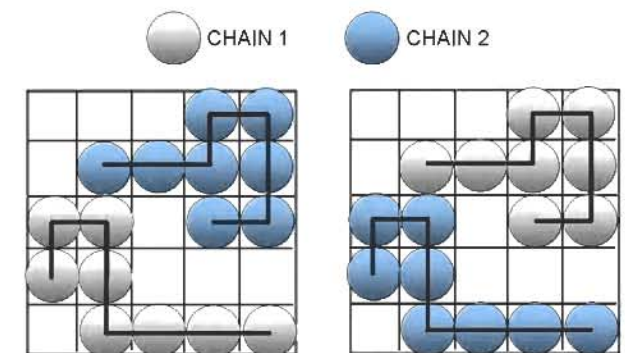
$$v_{subsequent} = \left( \frac{z-1}{M} \right)^{n_p(N-1)} \frac{(M-n_p)!}{(M-Nn_p)!}$$

- Putting these two configuration counts together, we have the total number of configurations for the collection of  $n_p$  chains of N segments each:

$$W = \frac{v_{first} v_{subsequent}}{n_p!}$$

$n_p!$

- The factor of  $n_p!$  Corrects for the over-counting since the polymer chains are indistinguishable, and we can't tell the difference between two configurations with the same polymer distributions but different chain identities:



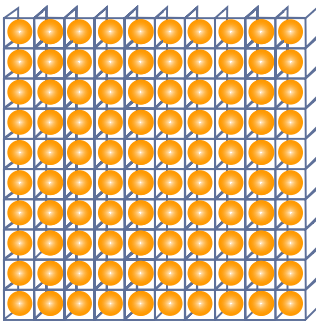
These two possible configurations are indistinguishable in the system.



- We are now ready to calculate the number of unique states for the unmixed and mixed states:

- UNMIXED STATE:

PURE SOLVENT:

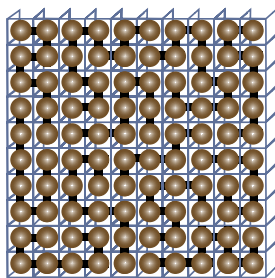


$$W_{\text{PURE SOLVENT}} = 1$$

$$\hookrightarrow S_{\text{PURE SOLVENT}} = k_b \ln W_{\text{PURE SOLV}} = 0$$

Figure by MIT OCW.

Pure Polymer



$$Nn_p = M$$

$$W_{\text{pure polymer}} = \left(\frac{z-1}{M}\right)^{n_p(N-1)} \frac{M!}{(M-Nn_p)n_p!} = z^{-1} n_p^{(N-1)} (Nn_p)!$$

$$S^{\text{pure polymer}} = k_b \ln W_{\text{pure polymer}}$$

Figure by MIT OCW.

MIXED STATE:

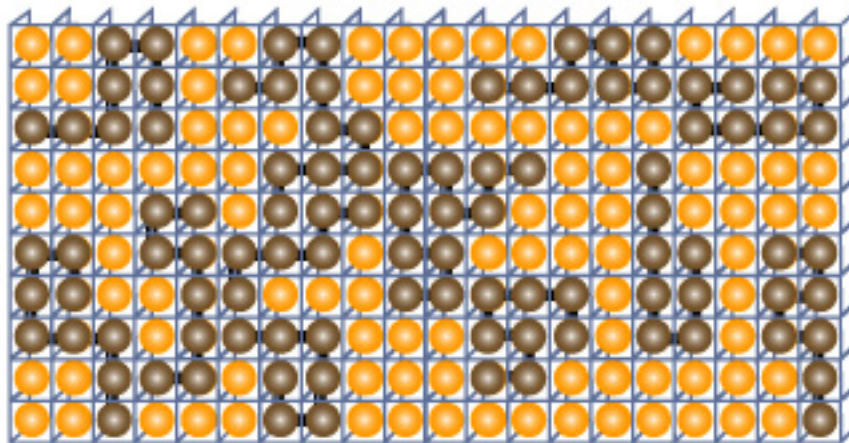


Figure by MIT OCW.

$$Nn_p + n_s = M$$

$$W_{\text{solution}} = \left(\frac{z-1}{M}\right)^{n_p(N-1)} \frac{M!}{(M-Nn_p)n_p!} = \left(\frac{z-1}{M}\right)^{n_p(N-1)} \frac{M!}{n_s! n_p!}$$

$$\Delta S^{\text{mix}} = S^{\text{solution}} - S^{\text{unmixed}} = S^{\text{solution}} - (S^{\text{pure solvent}} + S^{\text{pure polymer}}) = S^{\text{solution}} - S^{\text{pure polymer}}$$

$$\Delta S^{\text{mix}} = k_b \ln \frac{W_{\text{solution}}}{W_{\text{pure polymer}}} = k_b \ln \left( \frac{\left(\frac{z-1}{M}\right)^{n_p(N-1)} \frac{M!}{n_s! n_p!}}{\left(\frac{z-1}{M}\right)^{n_p(N-1)} \frac{M!}{(Nn_p)! n_p!}} \right) = k_b \ln \left( \frac{(Nn_p)^{n_p(N-1)}}{M} \frac{M!}{(Nn_p)! n_s!} \right)$$

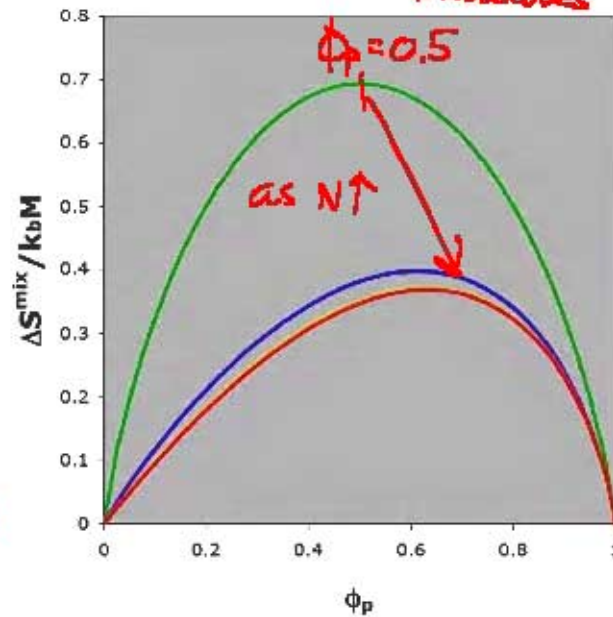
- Applying Stirling's approximation:  $\ln x! \approx x \ln x - x$ , ... arriving at a final result:

$$\therefore \Delta S^{mix} = -k_b [n_s \ln \phi_s + n_p \ln \phi_p]$$

COMPARE TO IDEAL SOLUTION!

$$\Delta S^{MIX, IDEAL} = -k_b [N_A \ln X_A + N_B \ln X_B]$$

# A, B MOLECULES



$\phi_i = X_i$

The energy of mixing in the Flory-Huggins model

- Molecules form bonds with one another, attract, and repel one another by electrostatic, polar, or Van der Waals interactions- changing the internal energy of the system.

USE MEAN FIELD LATTICE MODEL  
 ↓  
 $\Delta H^{mix}$   
 DESCRIBES REGULAR SOLUTION BEHAVIOR!

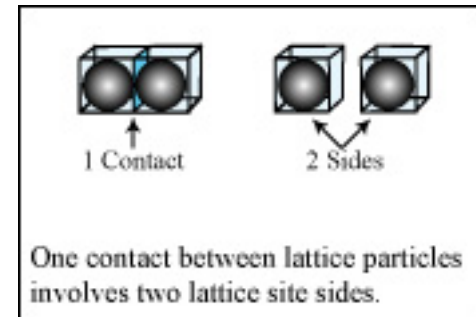
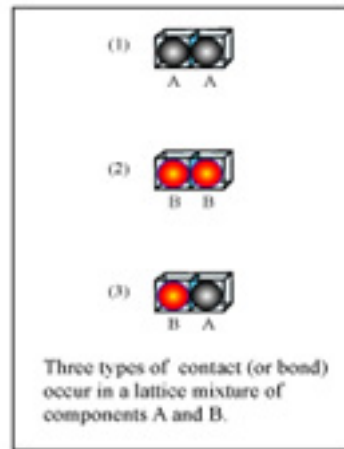


Figure by MIT OCW.

Figure by MIT OCW.

- To account for these interactions in a lattice model, we need to first calculate the enthalpy of mixing. At constant pressure, the enthalpy of mixing will be related to the internal energy of mixing in the following way:

$$H \equiv U + PV \quad \text{CONST. } P$$

$$\Delta H^{mix} = \Delta U^{mix} + P \Delta V^{mix} \approx \Delta U^{mix}$$

$\Delta V^{mix} \approx 0$

- We will assume in the lattice model that the change of volume on mixing is zero- thus the second term drops out and  $\Delta H^{mix} = \Delta U^{mix}$ .
- We can directly formulate an expression for the energy of mixing in the model:

$$\Delta U^{mix} = U_{SOLUTION} - U_{UNMIXED}$$

- The internal energy of the solution is obtained by considering the contacts between molecules on the lattice:

$$m_{ij} = \# \text{ i-j contacts}$$

$$\omega_{ij} = \text{energy per i-j contact}$$

$$U^{solution} = \sum (\# \text{ contacts}) \left( \frac{\text{energy}}{\text{contact}} \right) = m_{pp} \omega_{pp} + m_{ss} \omega_{ss} + m_{ps} \omega_{ps}$$

POLYMER-POLYMER CONTACTS      SOLVENT CONTACTS

- The total number of contacts made by *polymer segments* or *solvent* molecules on the lattice can be related to the coordination number and the contact numbers  $m_{ij}$ :

total solvent contacts:



$$n_s z = 2m_{ss} + m_{ps} \rightarrow m_{ss} = \frac{z n_s - m_{ps}}{2}$$

total polymer contacts:



$$N n_p z = 2m_{pp} + m_{ps} \rightarrow m_{pp} = \frac{z N n_p - m_{ps}}{2}$$

- Substituting into the expression for the internal energy of the solution, we have:

$$U^{solution} = m_{pp} \omega_{pp} + m_{ss} \omega_{ss} + m_{ps} \omega_{ps}$$

$$U^{solution} = \left( \frac{z N n_p - m_{ps}}{2} \right) \omega_{pp} + \left( \frac{z n_s - m_{ps}}{2} \right) \omega_{ss} + \omega_{ps} m_{ps}$$

$$U^{solution} = \frac{z N n_p}{2} \omega_{pp} + \frac{z n_s}{2} \omega_{ss} + \left( \omega_{ps} - \left( \frac{\omega_{pp} + \omega_{ss}}{2} \right) \right) m_{ps}$$

- The internal energy of the unmixed state has a simpler form:

$$U^{unmixed} = \underbrace{\left( \# \text{ s-s contacts} \right)}_{\text{PURE SOLVENT}} \omega_{ss} + \underbrace{\left( \# \text{ p-p contacts} \right)}_{\text{PURE POLYMER}} \omega_{pp}$$

$$= \left( \frac{z n_s}{2} \right) \omega_{ss} + \left( \frac{z N n_p}{2} \right) \omega_{pp}$$

- This leads us finally to an expression for the energy of mixing:

$$\Delta U^{mix} = U^{solution} - U^{unmixed} = \underbrace{\left( \omega_{ps} - \left( \frac{\omega_{ss} + \omega_{pp}}{2} \right) \right)}_{\text{ENERGY PER CONTACT!}} m_{ps}$$



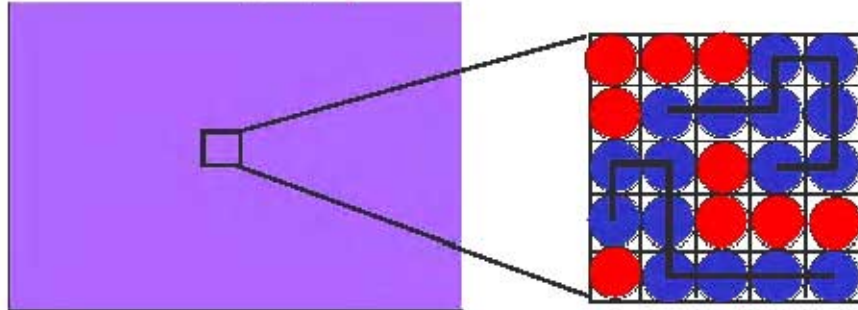
## The Bragg-Williams approximation

WHAT IS  $m_{ps}$ ?

- Our next step is to simplify the last equation- how can we determine  $m_{ps}$  from our model?
  - One way to achieve this is to make the approximation that the *polymer* and *solvent* molecules mix randomly on the lattice. In that case, all we need to do to find  $m_{ps}$  is calculate the probability of P-S contacts for random mixing. This is called the **mean field approximation** or the **Bragg-Williams approximation**. Note that we have already implicitly invoked this approximation in our excluded volume correction in the calculation of the entropy of mixing.

## MEAN FIELD APPROXIMATION!

$$\chi_s = 0.32$$



$$\chi_s = 0.32$$

LOCALITY: PROBABILITY OF FINDING A SOLVENT MOLECULE ON A GIVEN SITE!

- The probability that a lattice site contains a *polymer* segment is:

PROBABILITY OF POLYMER SEG. =  $\phi_p = \frac{N_p N}{M}$

- For a single *solvent* molecule, the average number of contacts with polymer segments is then:

# CONTACTS w/polymer =  $z \left( \frac{N_p N}{M} \right)$

- Thus, the total number of P-S contacts is:

$$\therefore m_{ps} = \left( \begin{array}{l} \text{\# SOLVENT MOLECULES} \\ \text{\# CONTACTS ON AVE. SOLVENT MOLECULE W/ POLYMER SEGMENTS} \end{array} \right) = v_s \left( \frac{z N_p N}{M} \right)$$

- This gives us a completely known expression for the energy of mixing in terms of the interaction energies:

$$\Delta U^{\text{mix}} = \left( \omega_{ps} - \left( \frac{\omega_{ss} + \omega_{pp}}{2} \right) \right) n_s \left( \frac{z n_p N}{M} \right)$$

- From this expression, the molar enthalpy of mixing is:

EQUIVALENT TO  $\Omega$   
FROM REGULAR SOLUTION  
MODEL

- We define the Flory-Huggins interaction parameter as:

$$\chi = \frac{z}{kT} \left( \omega_{ps} - \left( \frac{\omega_{ss} + \omega_{pp}}{2} \right) \right) \quad \text{UNITLESS ENERGY}$$

- $\chi$  is a unitless equivalent of  $\Omega$ , the interaction parameter from the regular solution model. Note that  $\Omega$  is thus directly related to the molecular energy of interaction between the two components of a binary system!
- The change in internal energy now reduces to:

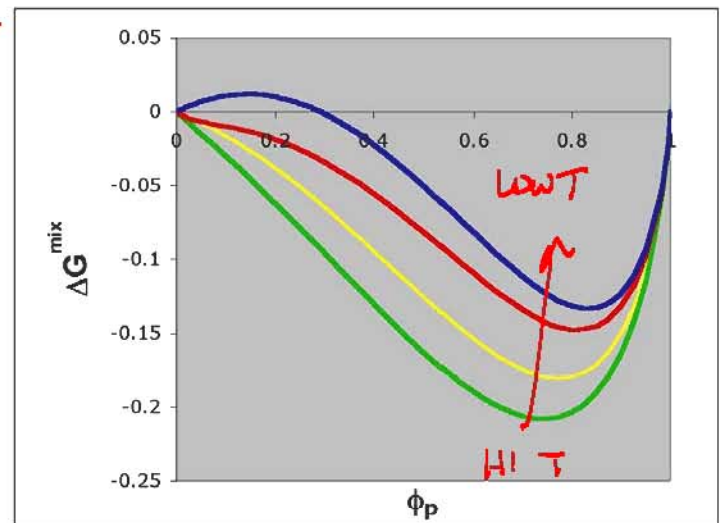
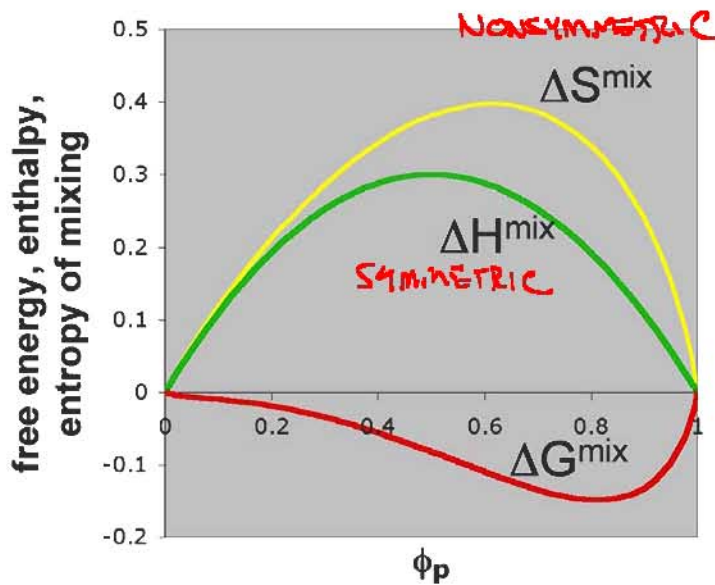
$$\Delta U^{\text{mix}} \Rightarrow \Delta H^{\text{mix}} = \chi kT \frac{n_s N_p}{M} = \chi k_B T \phi_s \phi_p M$$

$\nearrow M$   
 # LATTICE SITES

## Predictions from the Flory-Huggins model

- Now, this expression, derived directly from our molecule-scale model, can be combined with the lattice model entropy of mixing derived above to provide the free energy of mixing for the regular solution:

$$\begin{aligned} \Delta G^{\text{mix}} &= \Delta H^{\text{mix}} - T\Delta S^{\text{mix}} \\ &= \chi kT \phi_s \phi_p m + k_b T \left[ n_s \ln \phi_s + n_p \ln \phi_p \right] \\ \downarrow \\ \frac{\Delta G^{\text{mix}}}{kTm} &= \underbrace{\chi \phi_s \phi_p}_{\text{ENTHALPY}} + \underbrace{\phi_s \ln \phi_s + \frac{\phi_p}{N} \ln \phi_p}_{\text{ENTROPY}} \\ \phi_p &= \frac{n_p N}{m} \end{aligned}$$



REGULAR SOLUTION-LIKE  
BEHAVIOR

- The Flory-Huggins model predicts major trends in the behavior of real polymer solutions:

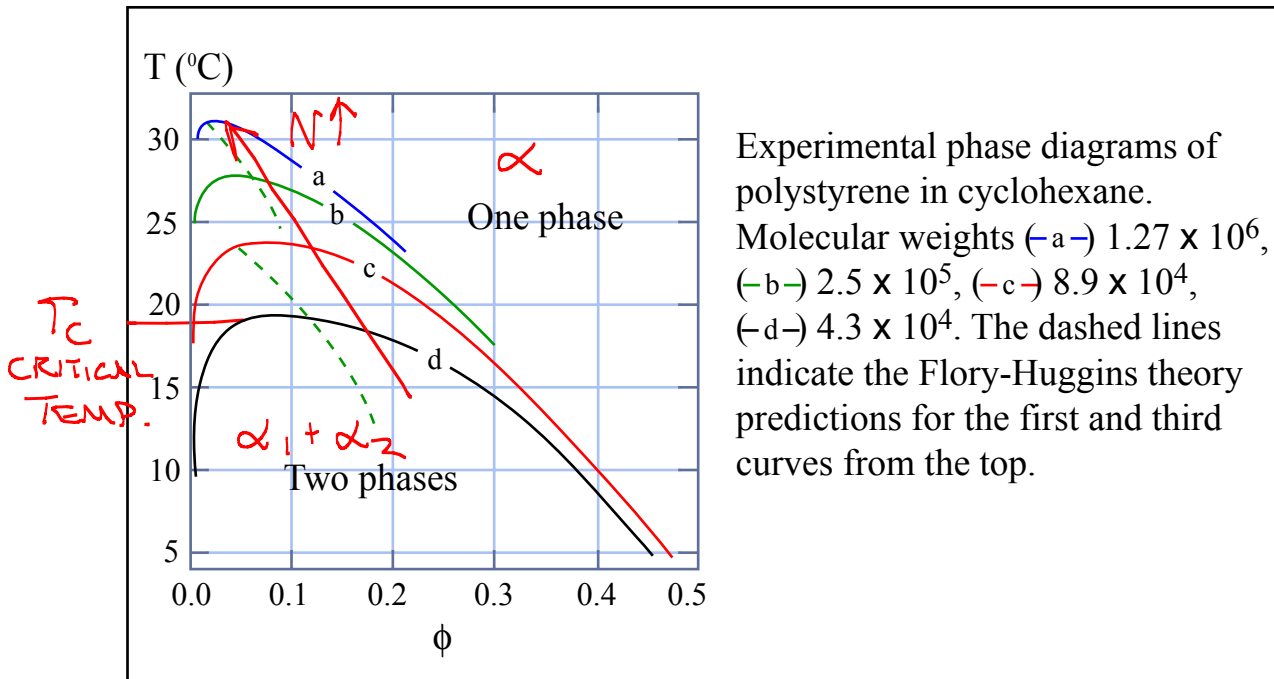
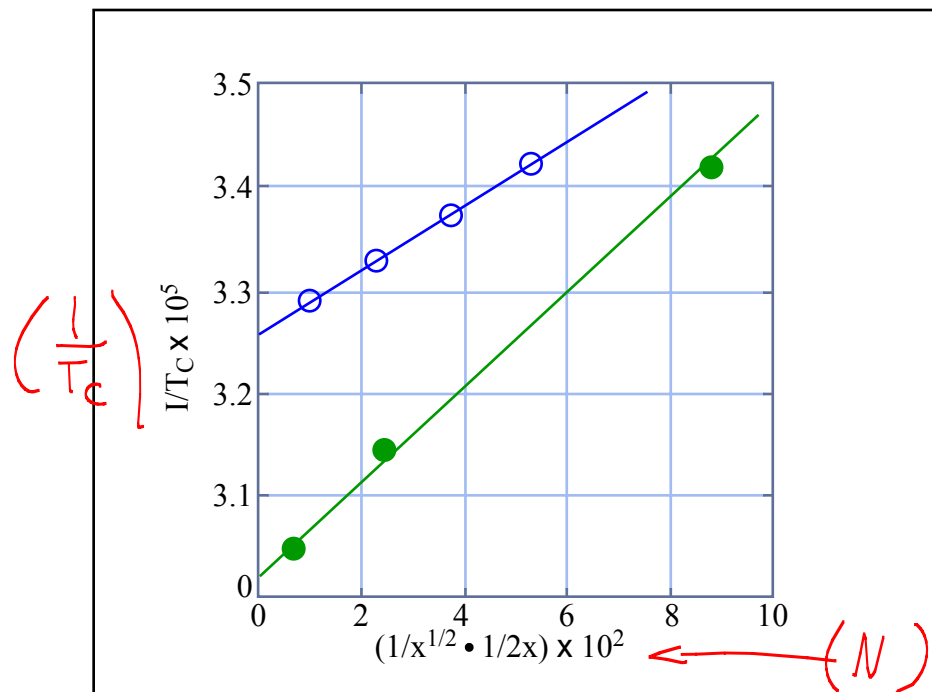


Figure by MIT OCW.



A plot of the reciprocal of the critical temperature against the molecular size function for polystyrene fractions in cyclohexane (○) and for polyisobutylene fractions in diisobutyl ketone (●).

Figure by MIT OCW.

- ...And this theory is used to predict new behavior in polymers in current research:

Image removed for copyright reasons.

Scanned image of article:

Gonzalez-Leon, Juan, Metin Acar, Ryu Sang-Woog, Anne-Valerie Ruzette, and Anne M. Mayes. "Low-temperature processing of 'baroplastics' by pressure-induced flow." *Nature* 426 (2003): 424-428.



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**References**

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1. Dill, K., and S. Bromberg. 2003. *Molecular Driving Forces*, New York.
2. Flory, P. J. 1953. *Principles of Polymer Chemistry*. Cornell University Press, Ithaca.